

Fracture toughness of architected joints involving crack instabilities

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Composite



+

Composite



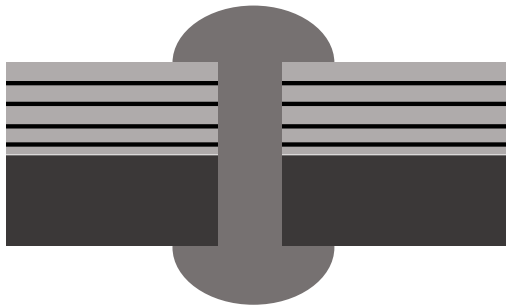
Metal (Ti, Al ...)



→ To bond

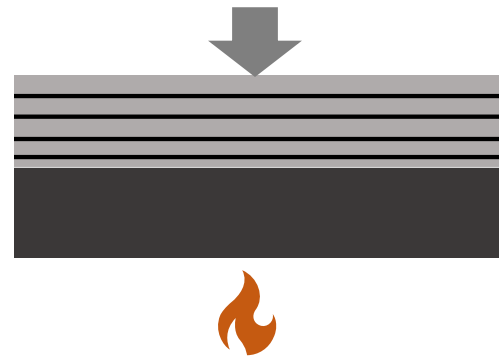
Joining techniques

Mechanical fastening



- ✗ Drilling holes
→ stress concentrations
→ ↓ structural integrity
- ✗ Additional weight
- ✓ Control of components

Thermoplastic welding



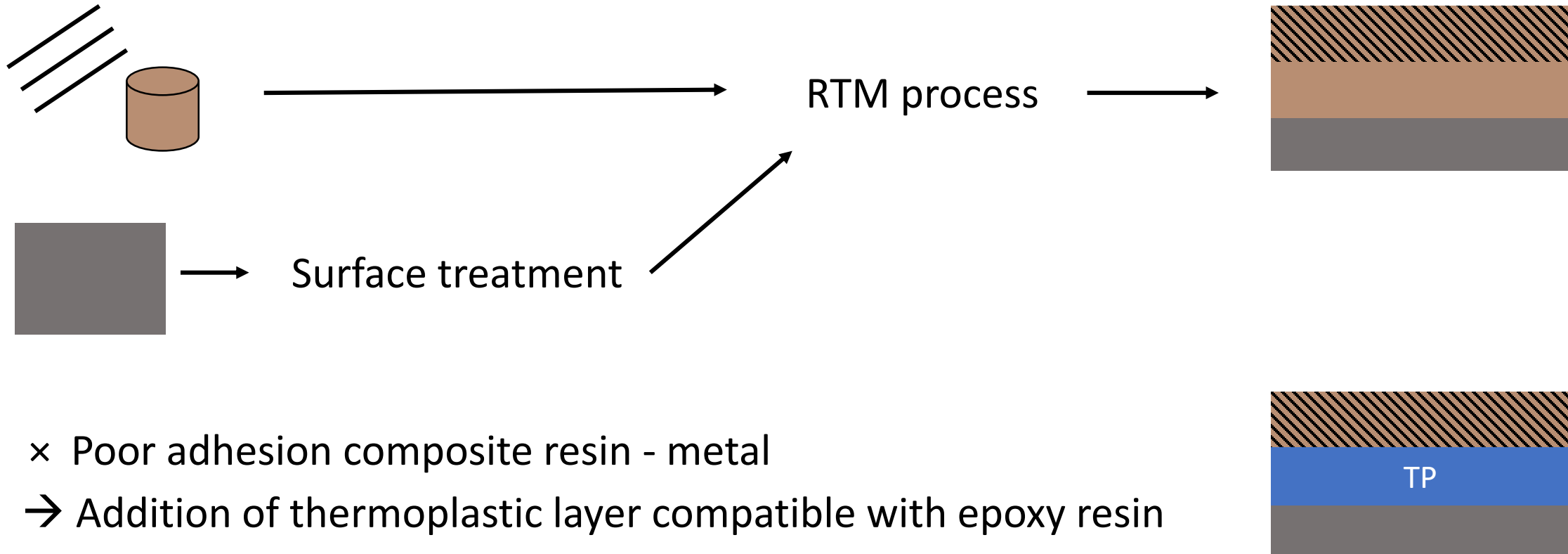
- ✗ Only for TP composites
→ TS = largest share
- ✓ High throughput

Adhesive bonding

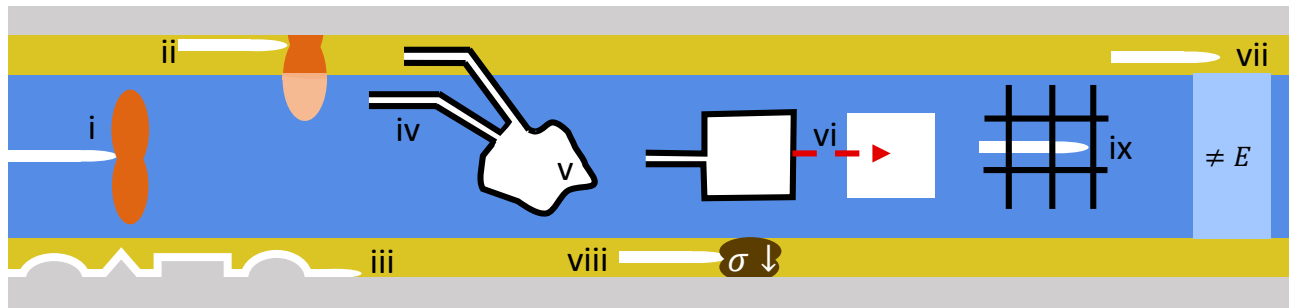


- ✓ Homogeneous stress transfer
- ✓ Lightweight
- ✗ Low G_{IC}
- ✗ Time- & labor-intensive
- ✗ Lack of non-destructive control

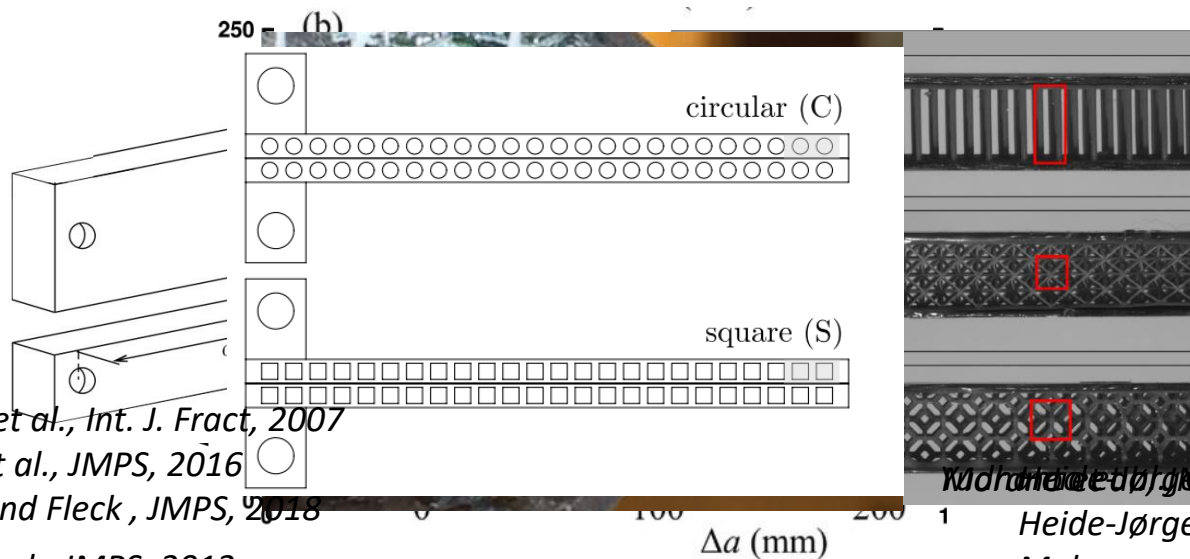
Time- and labor-intensive process → Integrated manufacturing = Cocuring



Low $G_{Ic} \rightarrow$ Toughening strategies



Plastic dissipation (i-ii)
Interface patterning (iii)
Crack deflection (iv)
Crack tip blunting (v)
Metamaterials (vi)
Bridging (ix)
Change of stiffness (vii-viii)



Zavattieri et al., *Int. J. Fract.*, 2007

Cordisco et al., *JMPS*, 2016

Maloney and Fleck, *JMPS*, 2018

Martiny et al., *JMPS*, 2012

Heide-Jørgensen et al., *JMPS*, 2020

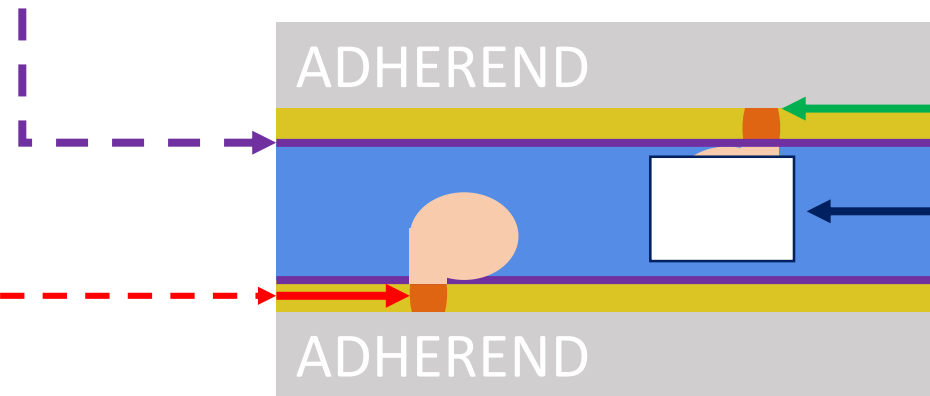
Wahne et al., *JMPS*, 2018; Birkhoff et al., *JMPS*, 2017; Heide-Jørgensen et al., *JMPS*, 2017

Heide-Jørgensen and Budzik, *JMPS*, 2018

Maloney and Fleck, *JMPS*, 2019

Tough architected joints + cocuring

- ✓ Bonding: tough interface TP-resin
 - Semi IPNs (PEI, PES, PSU ...)
 - Reaction of functional groups (PA)



S1: Composite interface = C-fibres
– resin interface

- No control
- $\uparrow G_{IC}$: plastic dissipation in TP if soft

S2: Resin layer

- Too thin for architecturing
- $\uparrow G_{IC}$: plastic dissipation in TP if soft

S3: TP film

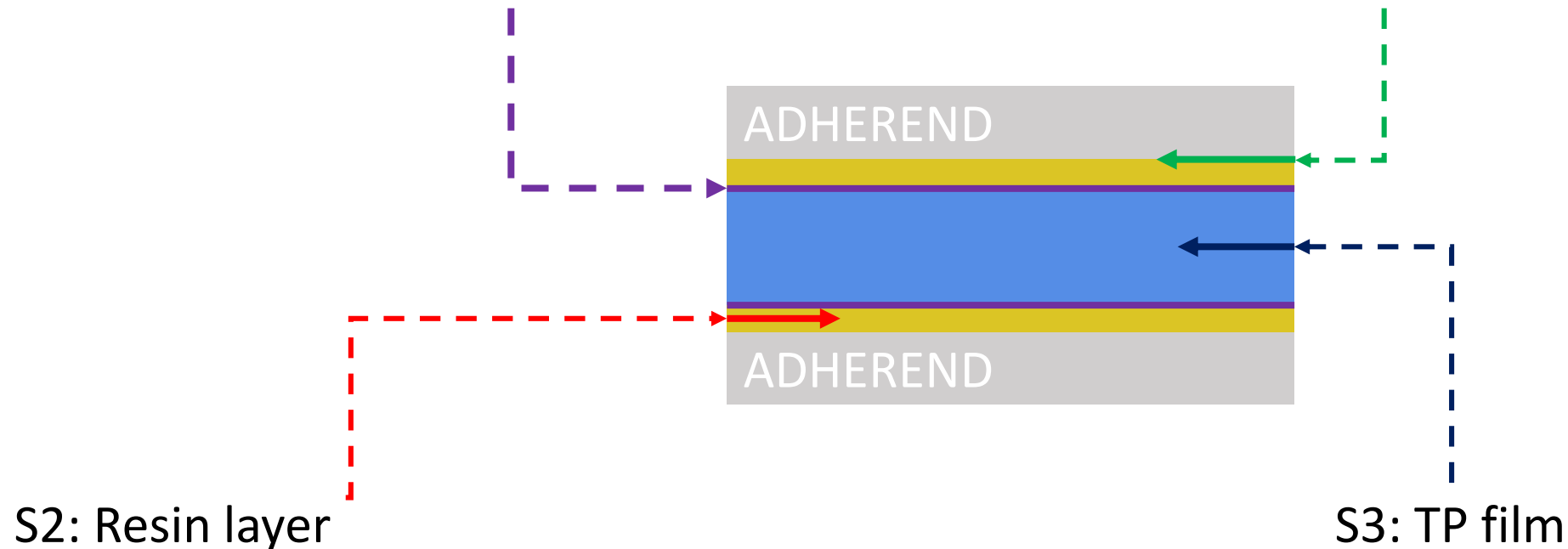
- Architecturing

Toughest

Tough architected joints + cocuring

✓ Bonding: tough interface TP-resin interface

S1: Composite interface = C-fibres
– resin interface

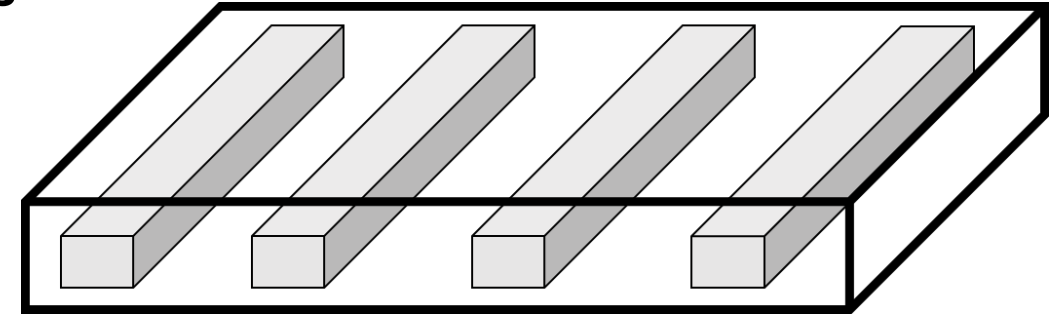


Dominant scenario = weakest

➔ deflection to trigger other mechanisms ➔ addition of cavities in the TP layer

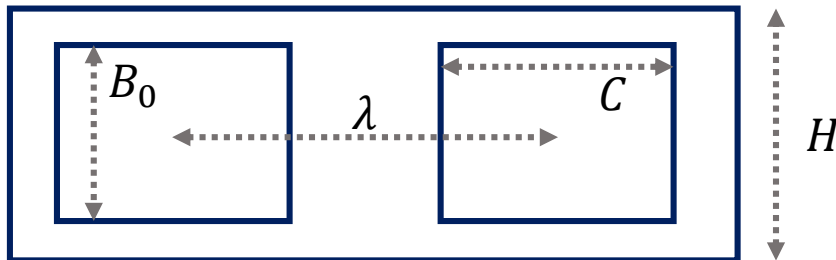
A thermoplastic film with some characteristics

- (1) Good interactions with the epoxy resin
- (2) Softer than the resin
- (3) Contains cavities
- (4) Easy to manufacture → 3D printing

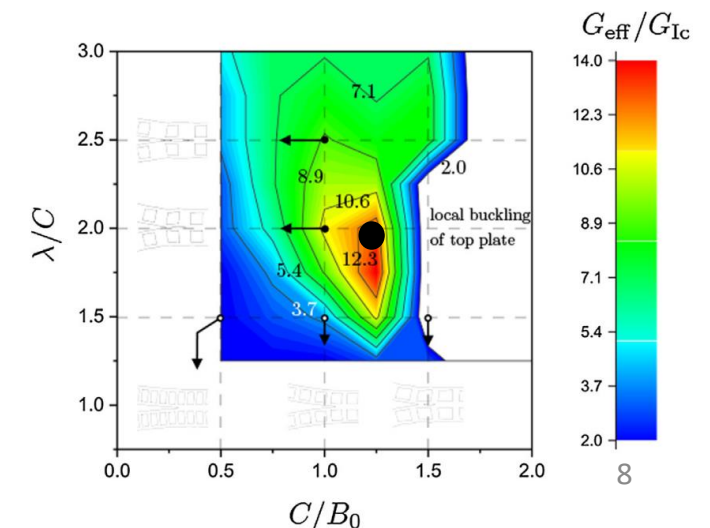
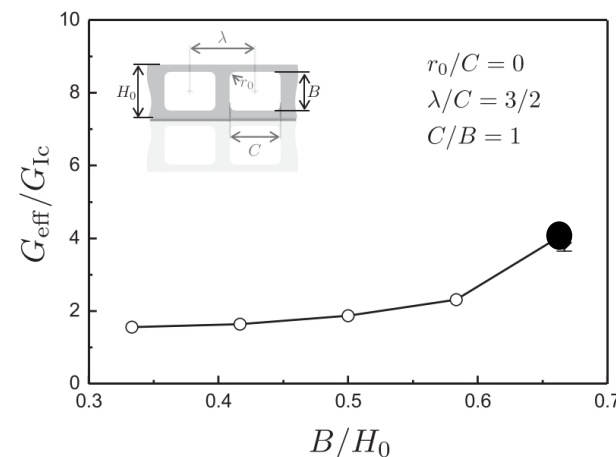


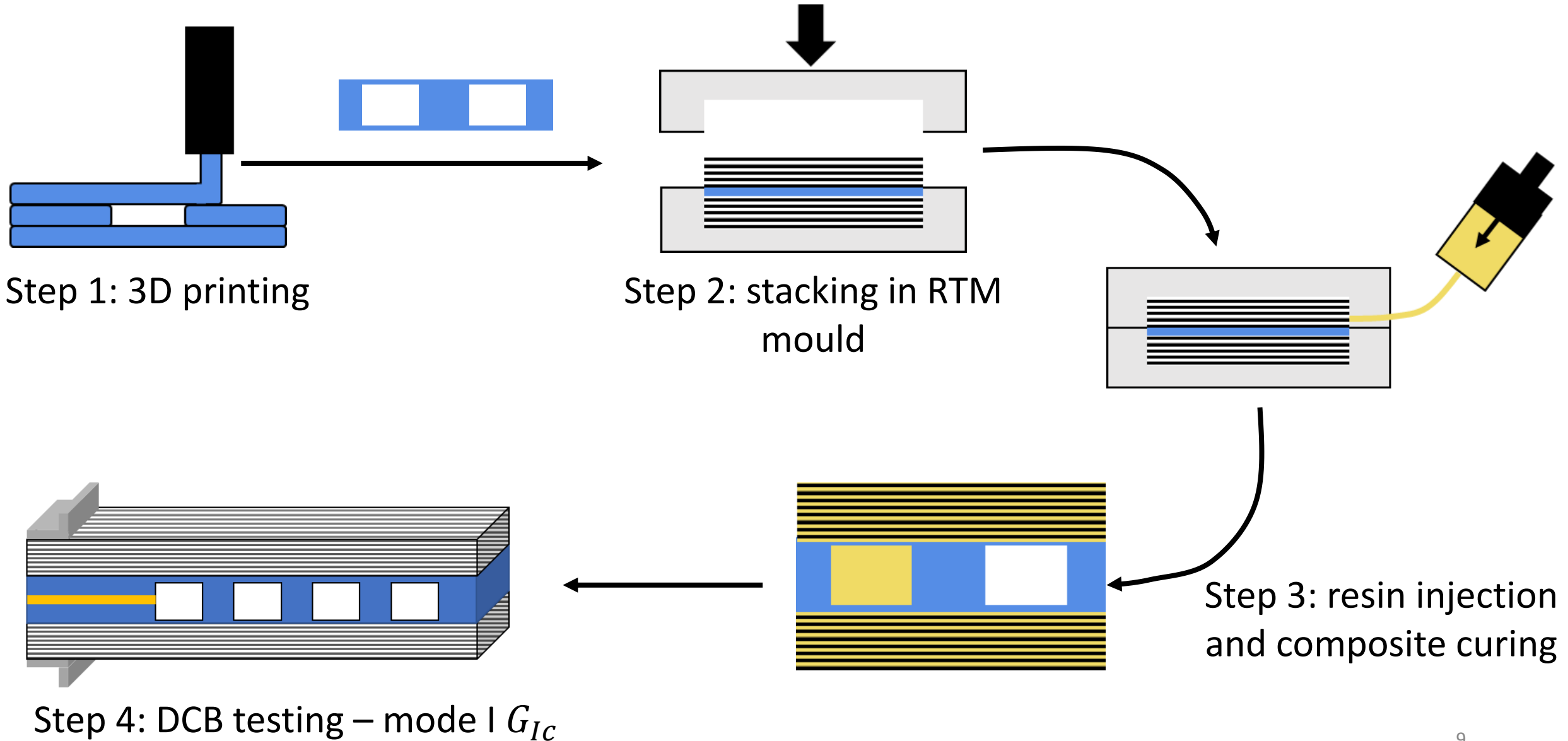
Dimensions of the cavities

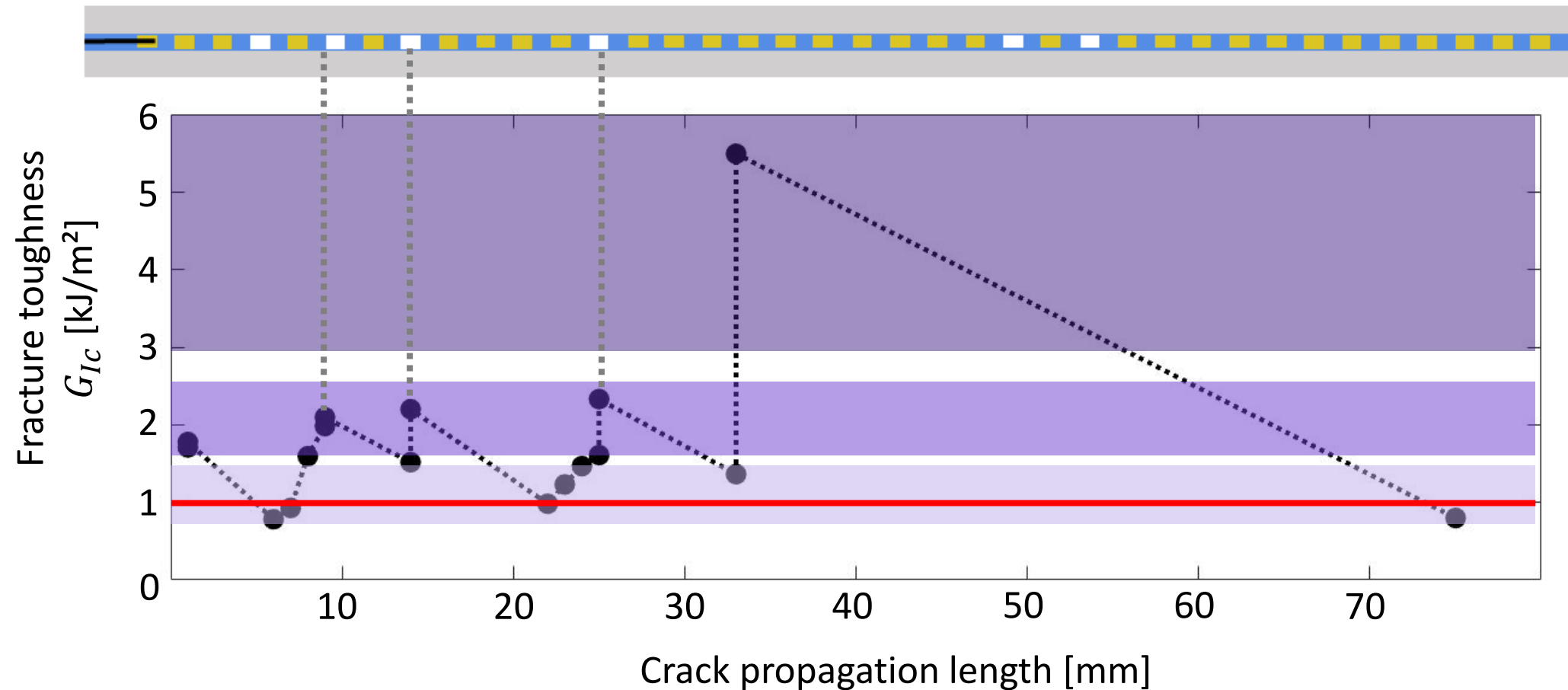
- (1) Manufacturing constraint: depth of RTM mold → film thickness of 1 mm
- (2) Dimensions of cavities: parametric study of Morano et al.



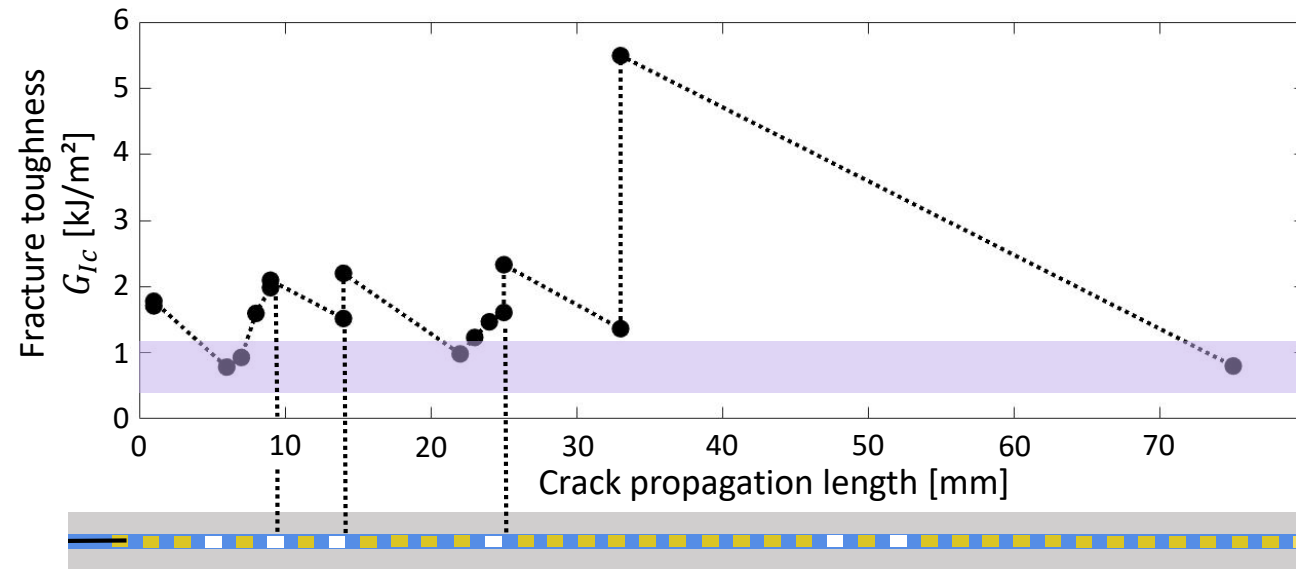
Morano et al., JMPS, 2020



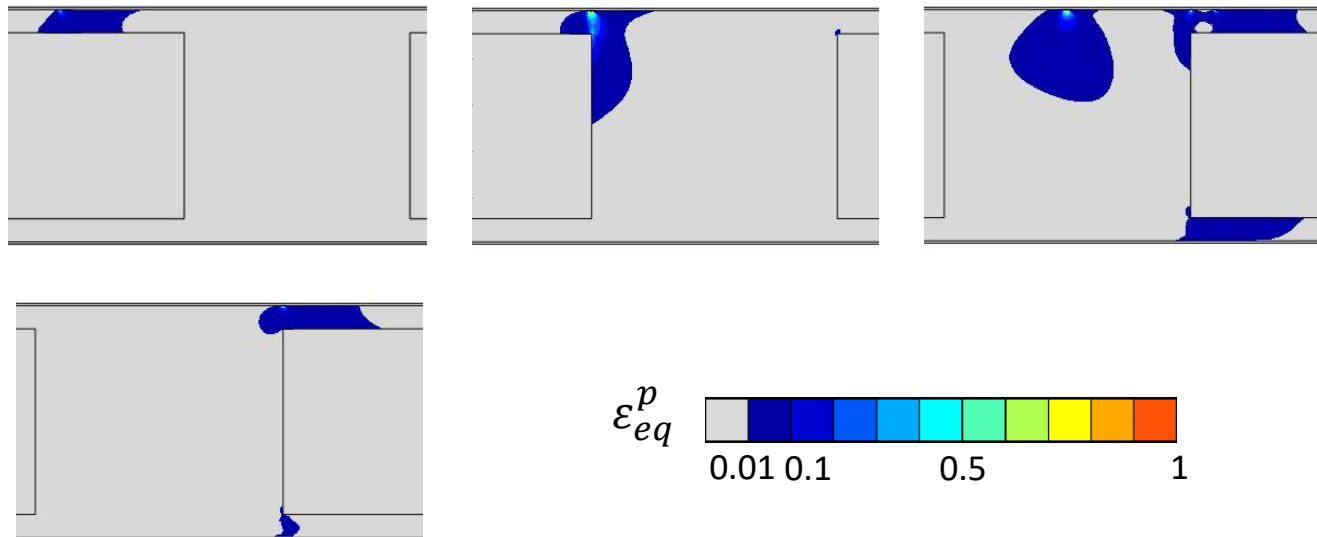
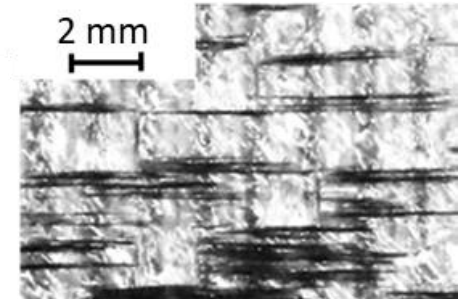
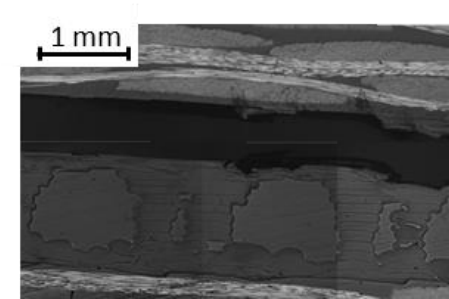
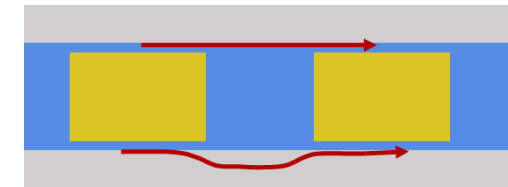




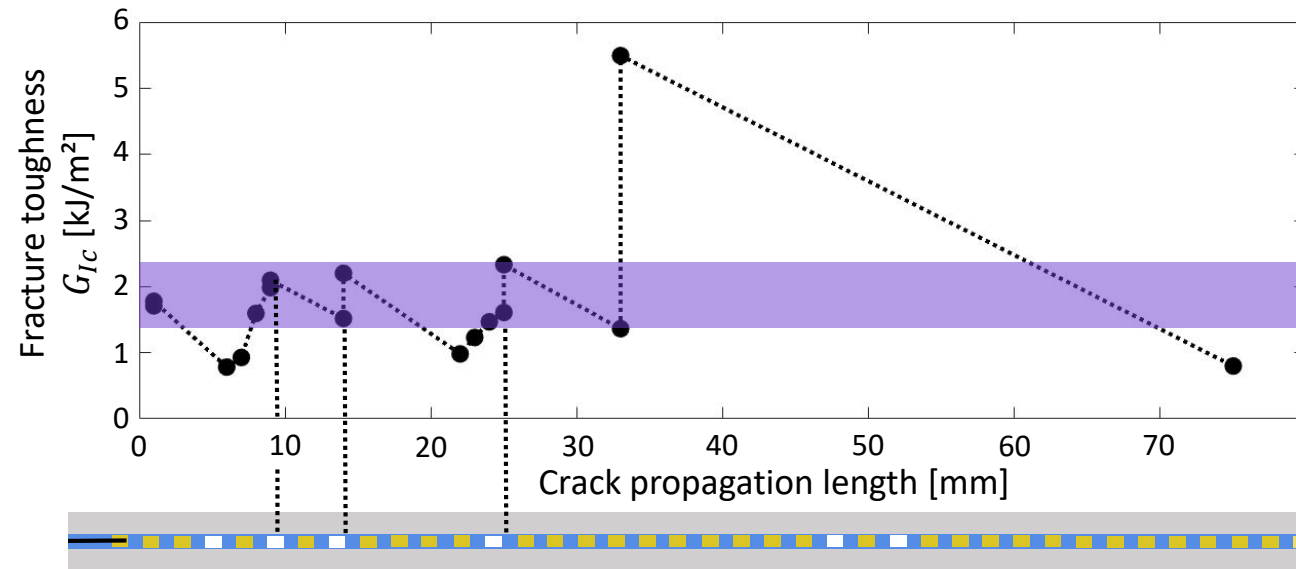
Stick-slip behavior → Identification of 3 G_{IC} regimes
→ Crack instabilities



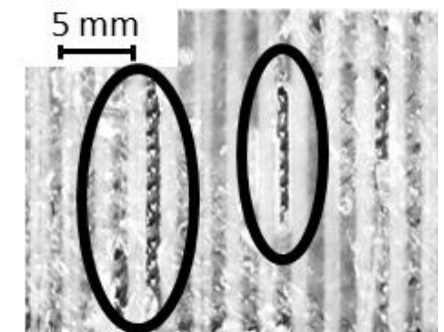
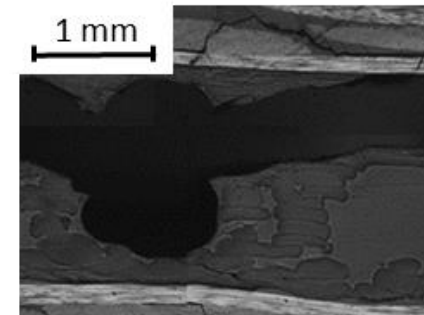
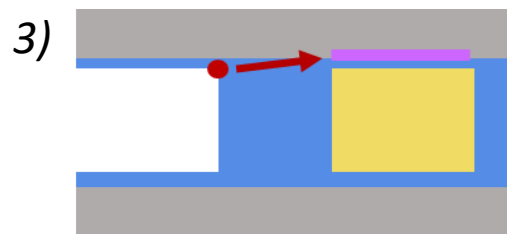
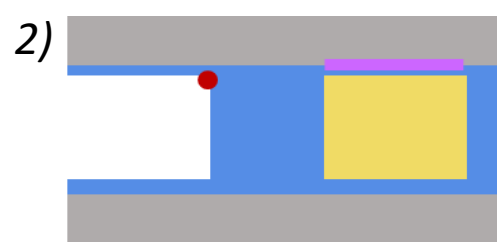
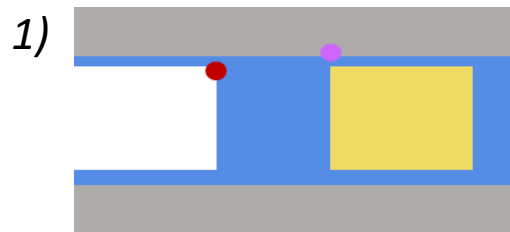
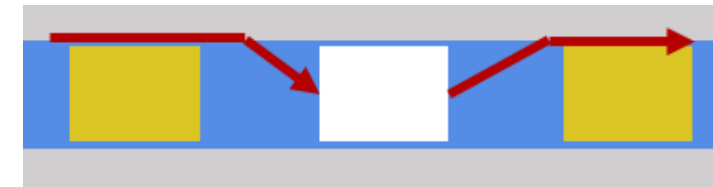
**Regime (1): propagation at composite-Nylon interface:
600-1200 J/m²**



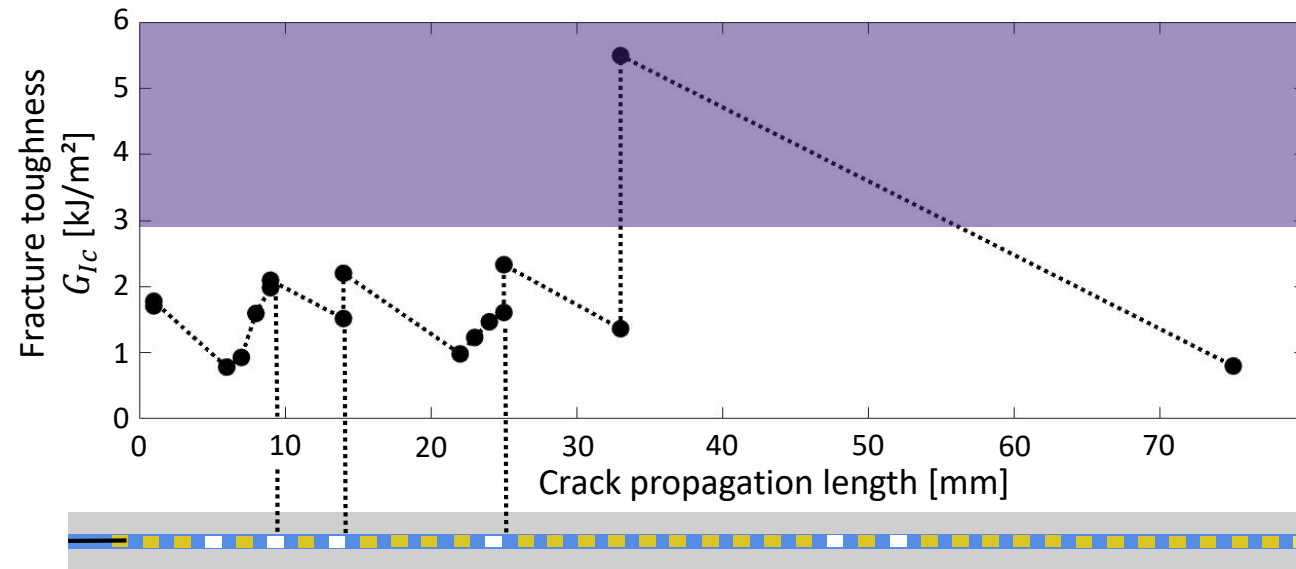
Architected film: dissipative layer
+ mechanical interlocking



**Regime (2): empty channel - deflection and re-initiation:
1500-2200 J/m²**



Empty channels: deflection +
propagation in tough TP



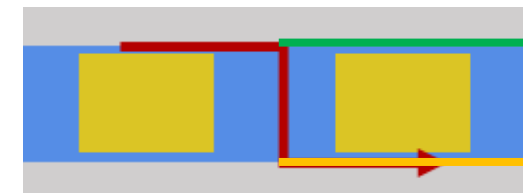
Controlling channel filling

→ different toughness regimes

→ **Triggering of different toughening mechanisms**

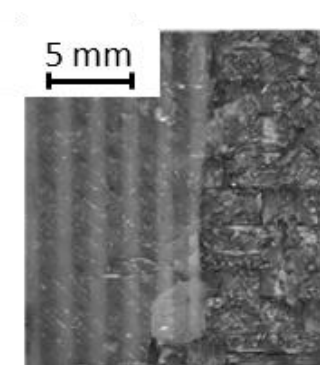
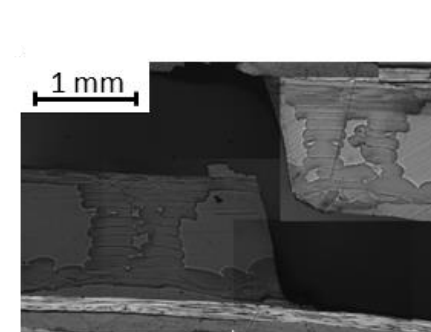
Change of regime associated with crack arrest or jumps

Regime (3): propagation through Nylon > 3000 J/m²



Strong

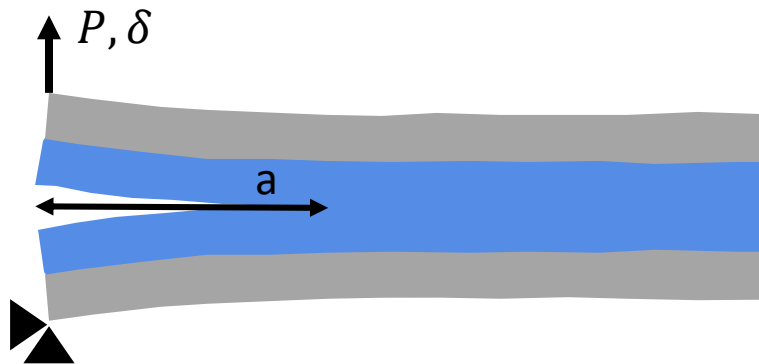
Weak



Local change in adherence?

Analytical model to consider the stick-slip behaviour

DCB $\rightarrow$ Euler-Bernoulli beams



$$G_{Ic} = \frac{P^2 a^2}{EBI} = \frac{3Eh^3 \delta^2}{16a^4}$$

Stick-slip behavior: triggering of different mechanisms

- 3 regimes with 3 different G_{Ic} : $G_{Ic}^1 < G_{Ic}^2 < G_{Ic}^3$
- a at which we change of regimes are known (experiment)

Three scenarios

1. Steady-state: $G_{IC} = ct$

$$P(\delta, G_i, a_i) = \left(\frac{EB^4 h^3 G_i}{27} \right)^{1/4} \frac{1}{\sqrt{\delta_i}}$$

2. Crack jump from a_i to a_{i+1} : $\delta_{i+1} = \delta_i$ and $G_i > G_{i+1}$

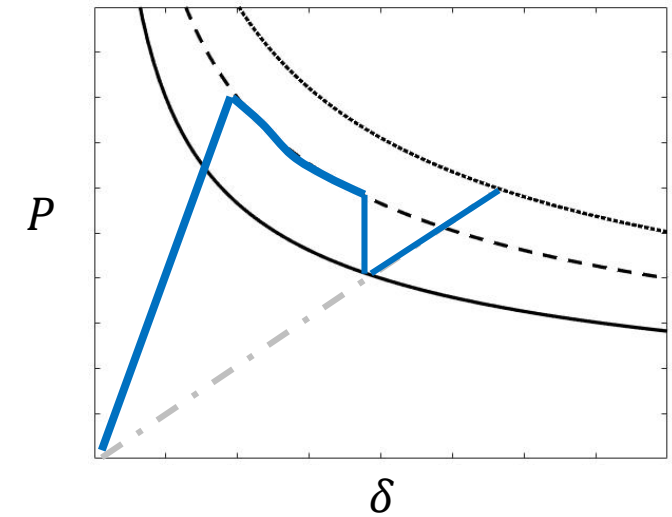
$$\Delta a = a_{i+1} - a_i = \left(\frac{3Eh^3}{16} \right)^{1/4} \sqrt{\delta_i} \left(\frac{1}{G_{i+1}^4} - \frac{1}{G_i^4} \right)$$

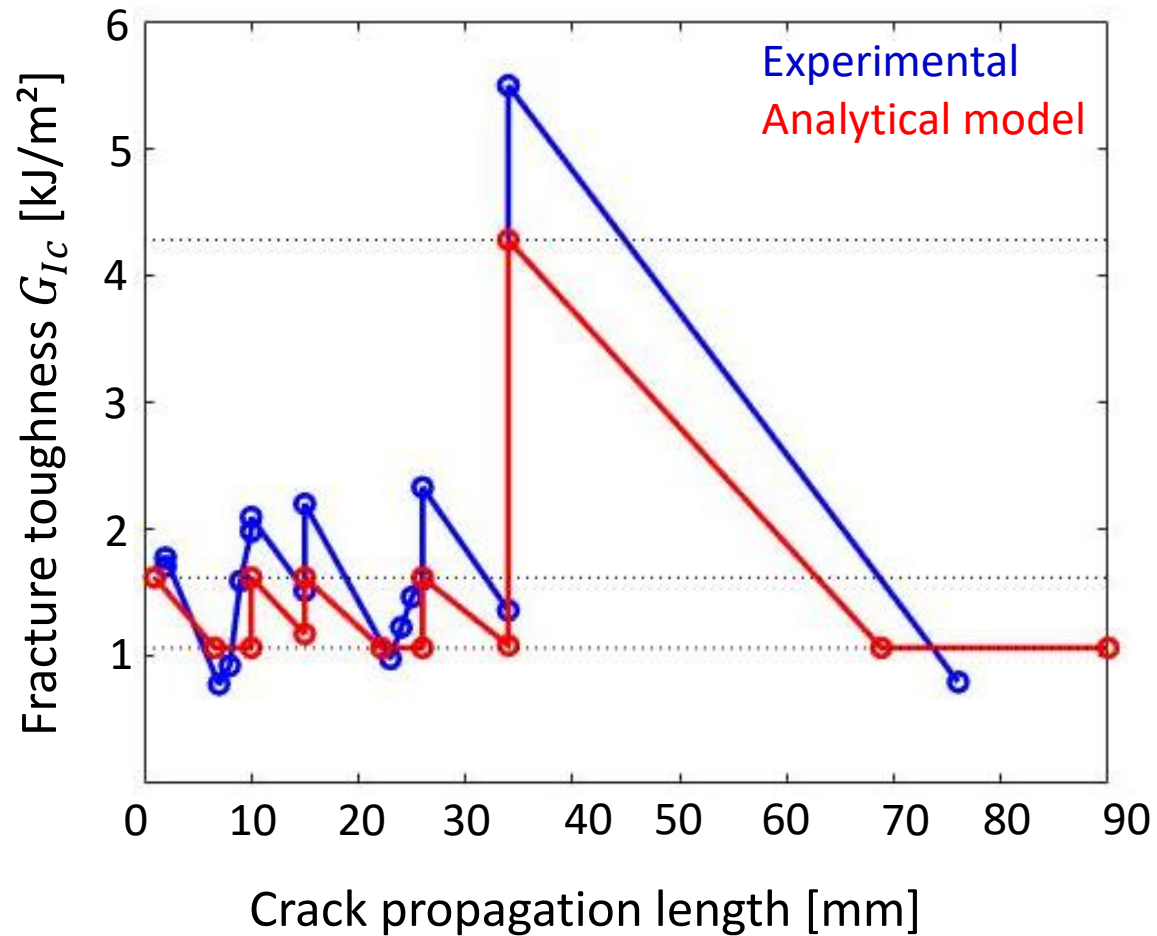
$$\Delta P = \left(\frac{EB^4 h^3}{27} \right)^{1/4} \frac{1}{\sqrt{\delta_i}} \left(G_{i+1}^{1/4} - G_i^{1/4} \right)$$

3. Crack arrest $a_i = a_{i+1}$ and $G_i < G_{i+1}$

$$\delta_{i+1} = \delta_i \sqrt{\frac{G_{i+1}}{G_i}}$$

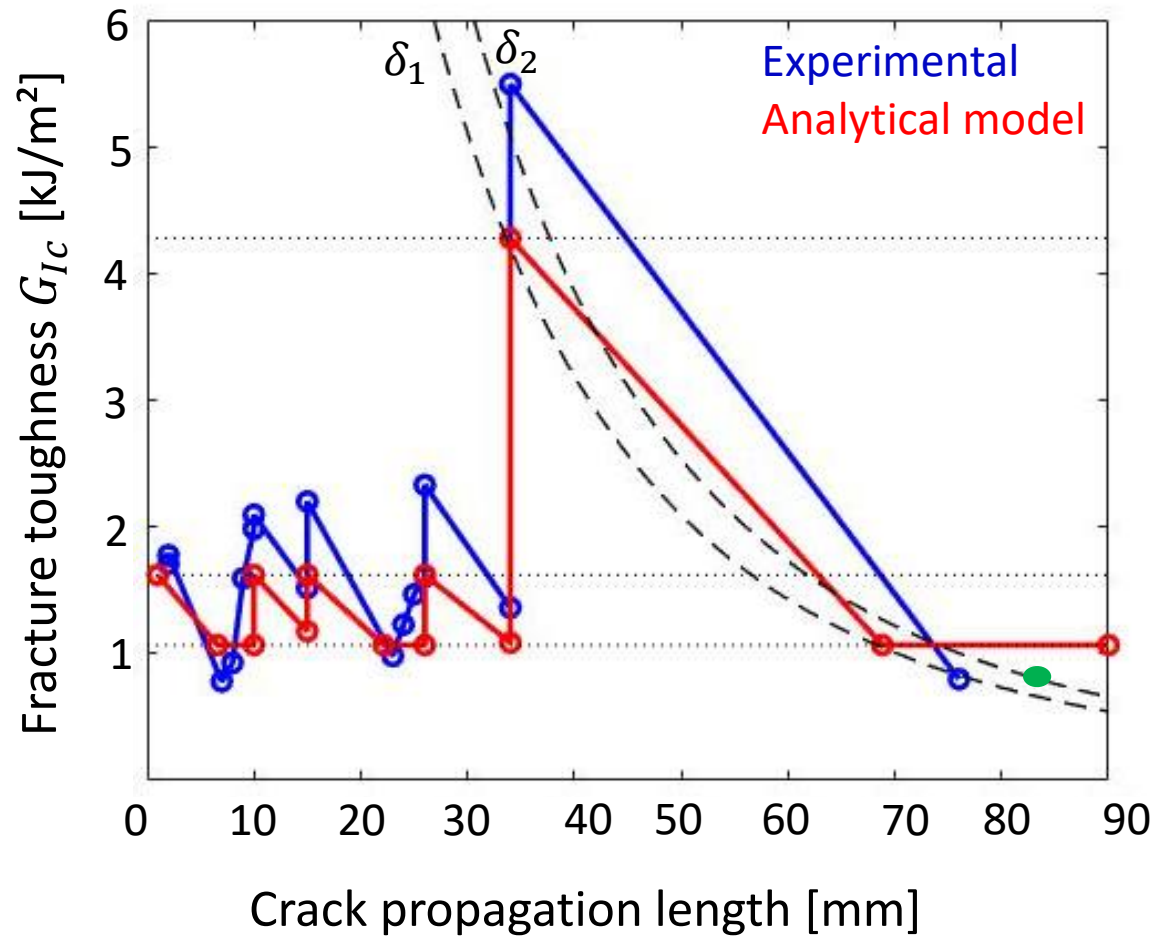
$$P_{i+1} = \left(\frac{EB^4 h^3 G_i}{27} \right)^{1/4} \sqrt{\frac{G_{i+1}}{\delta_i}}$$





Model

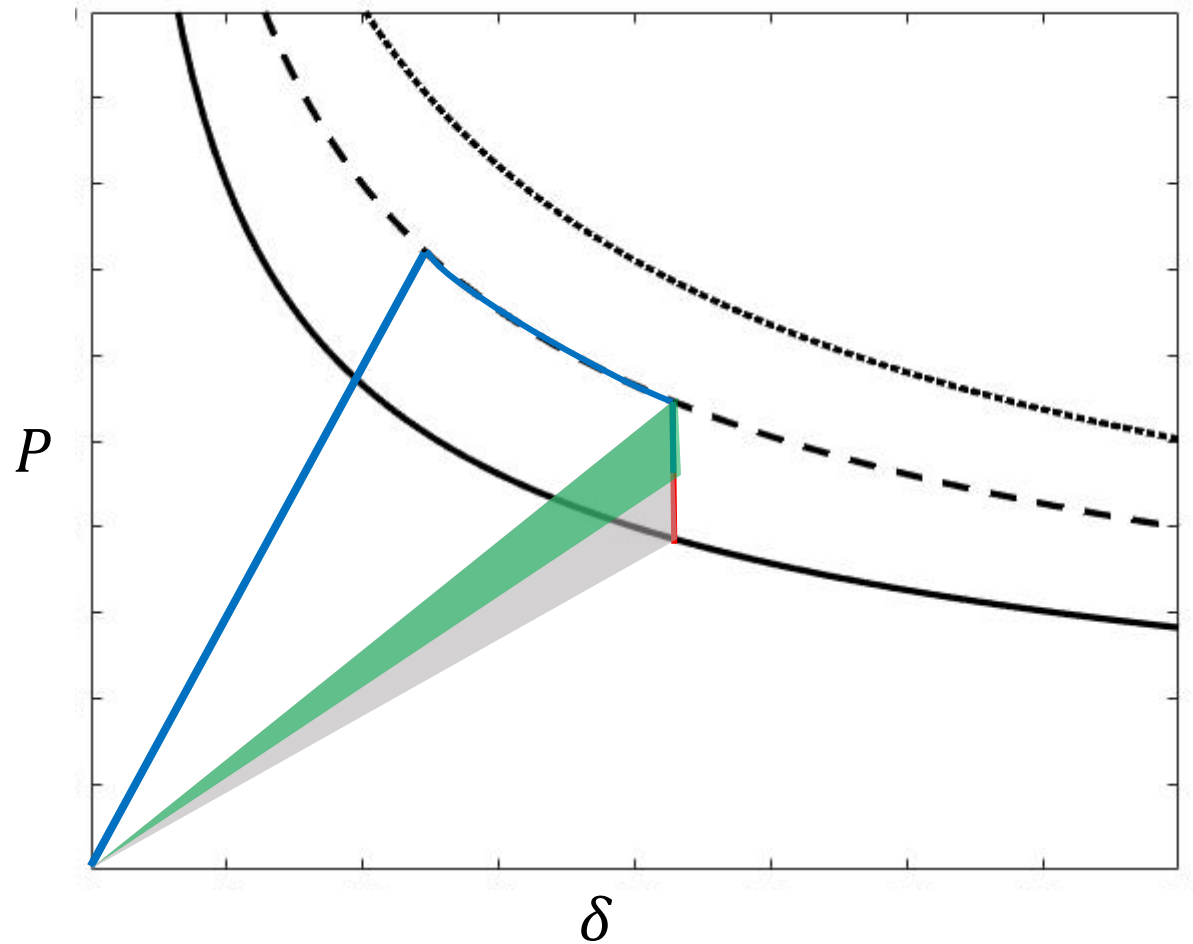
- Average fracture toughness for each regime
- Crack position for each event
- ✓ Capture the general trend of the R-curve



Model

- Average fracture toughness for each regime
- Crack position for each event
- ✓ Capture the general trend of the R-curve
- × Overestimation of crack jump length

$$G_{Ic}(\delta) = \frac{3Eh^3\delta^2}{16a^4}$$



Equilibrium equation

$$dW = -\frac{1}{2}\Delta P\delta = G_c^1\Delta aB$$

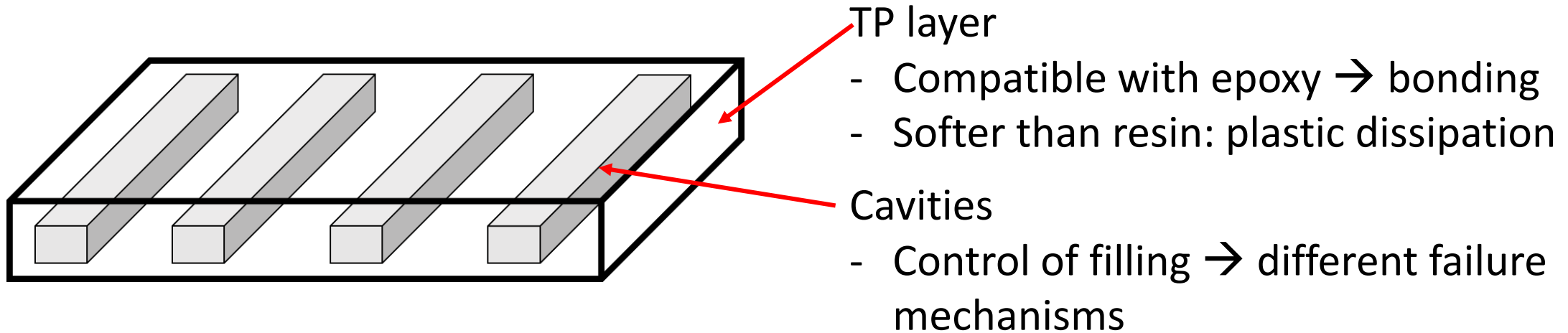
Modified equilibrium equation

$$dW = -\frac{1}{2}\Delta P\delta = G_c^1\Delta aB - \text{others}$$

- Dissipation ahead of crack tip?
- Influence of kinetic energy?

Simple EB model fails to correctly predict the joint behavior

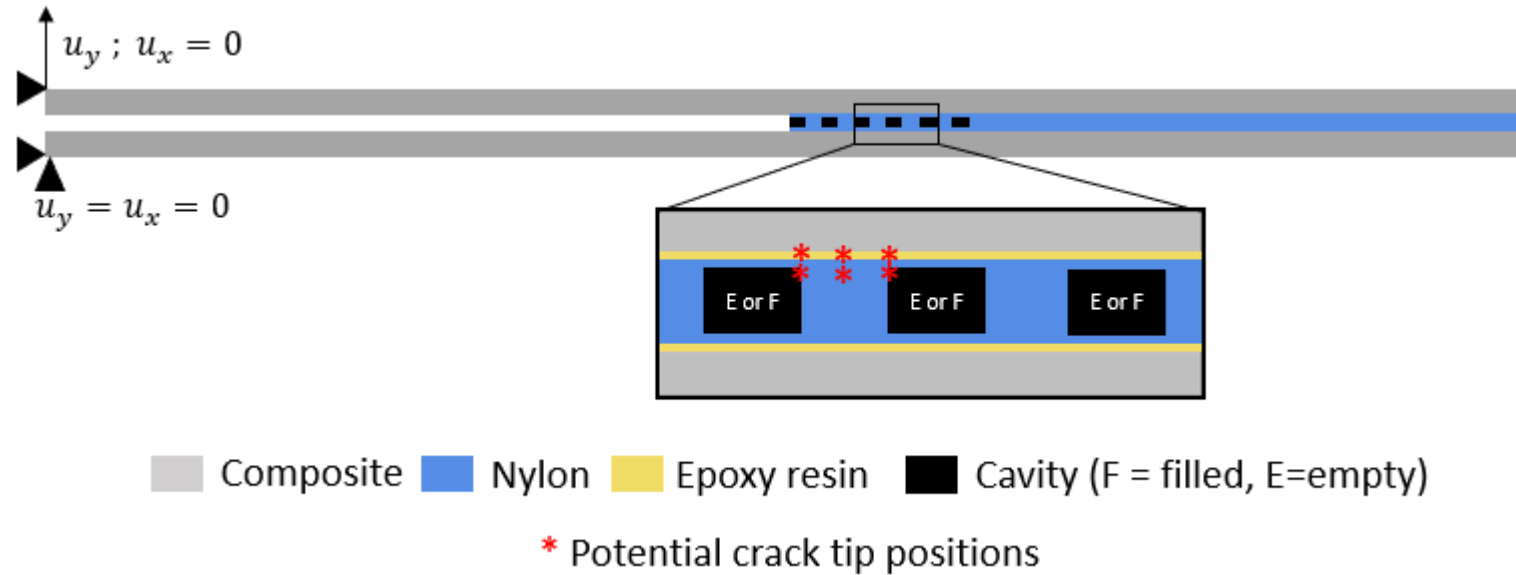
Initial objective: design of tough joints for single step bonding process



G_{Ic} from 600 J/m² up to 5500 J/m²

Stick-slip behavior → not accurately captured by simple analytical model based on EB beam theory

→ Need to consider dissipation phenomenon (to investigate)

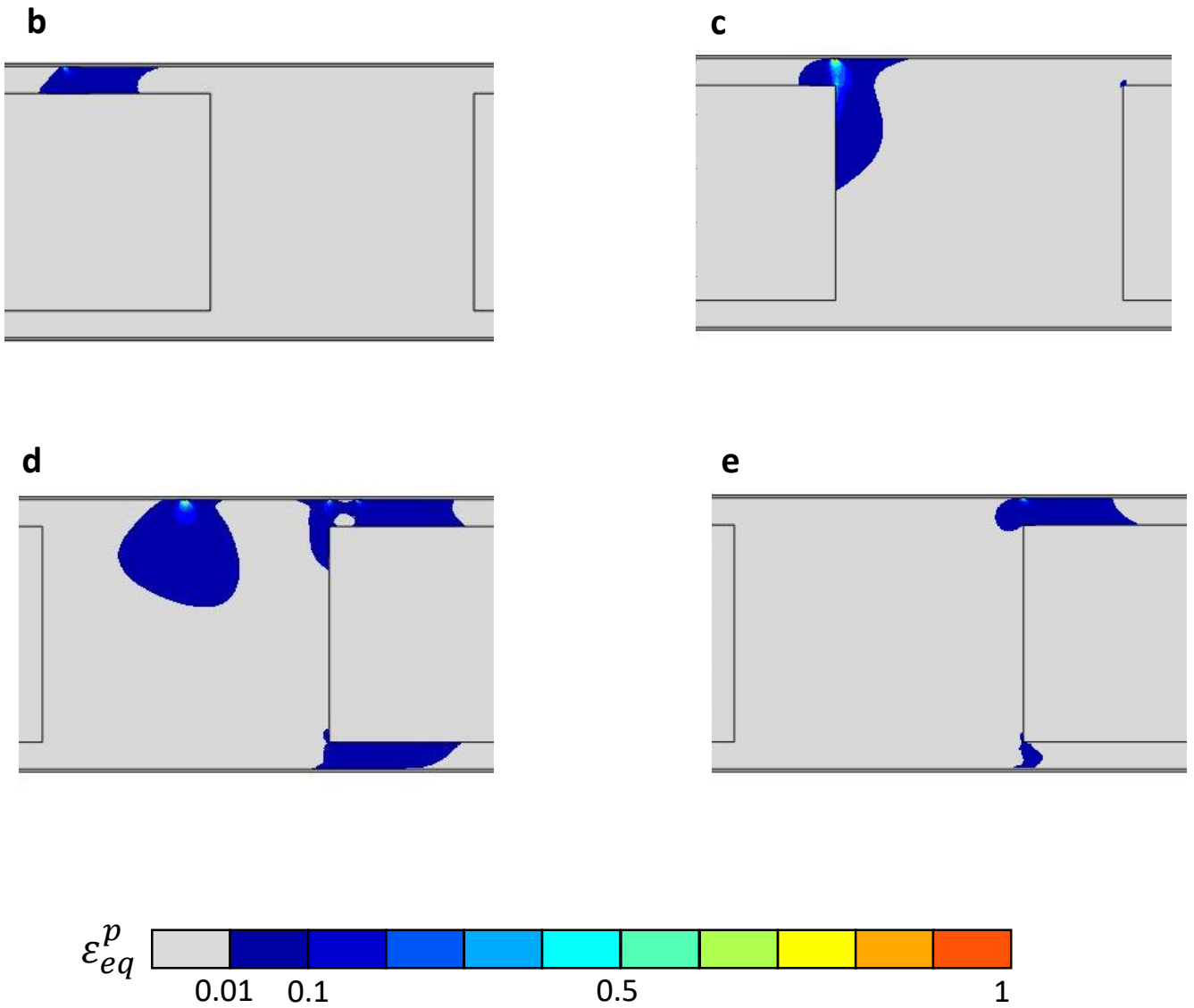
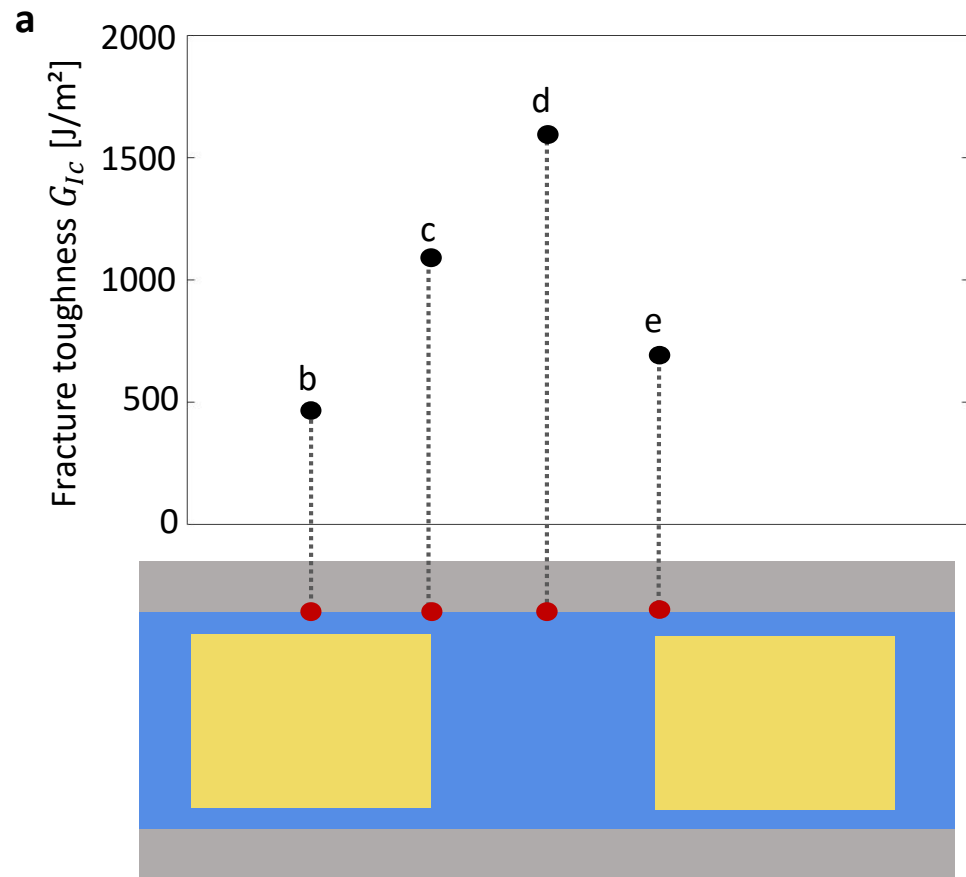


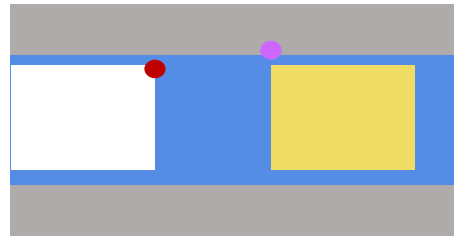
Material constitutive laws

- Composite: $E_{in} = 70 \text{ GPa}$ and $E_{out} = 3 \text{ GPa}$
- Resin: Drucker-Prager model for RTM6 [Morelle et al., Mech. Time-Dependent Mat., 2017]
- Nylon: properties from tensile tests

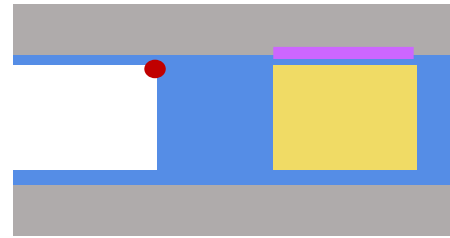
Failure criterion: σ_c at d_c

- RTM6: 300 MPa, 10 μm
- Interface: 50 MPa, 100 μm
- Nylon: 50 MPa, 200 μm

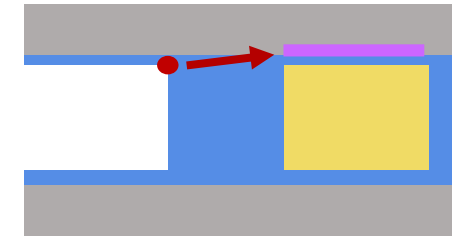




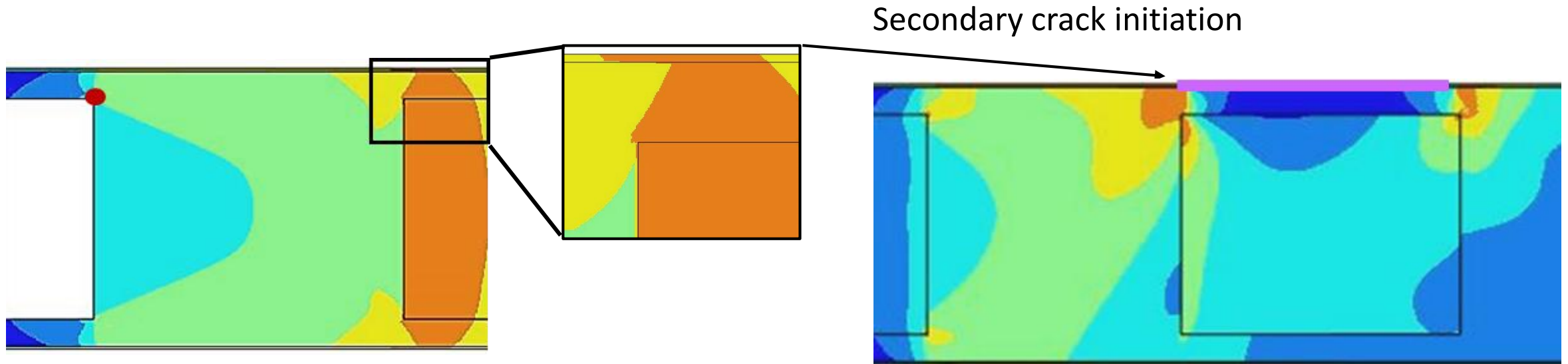
1) Secondary crack initiation



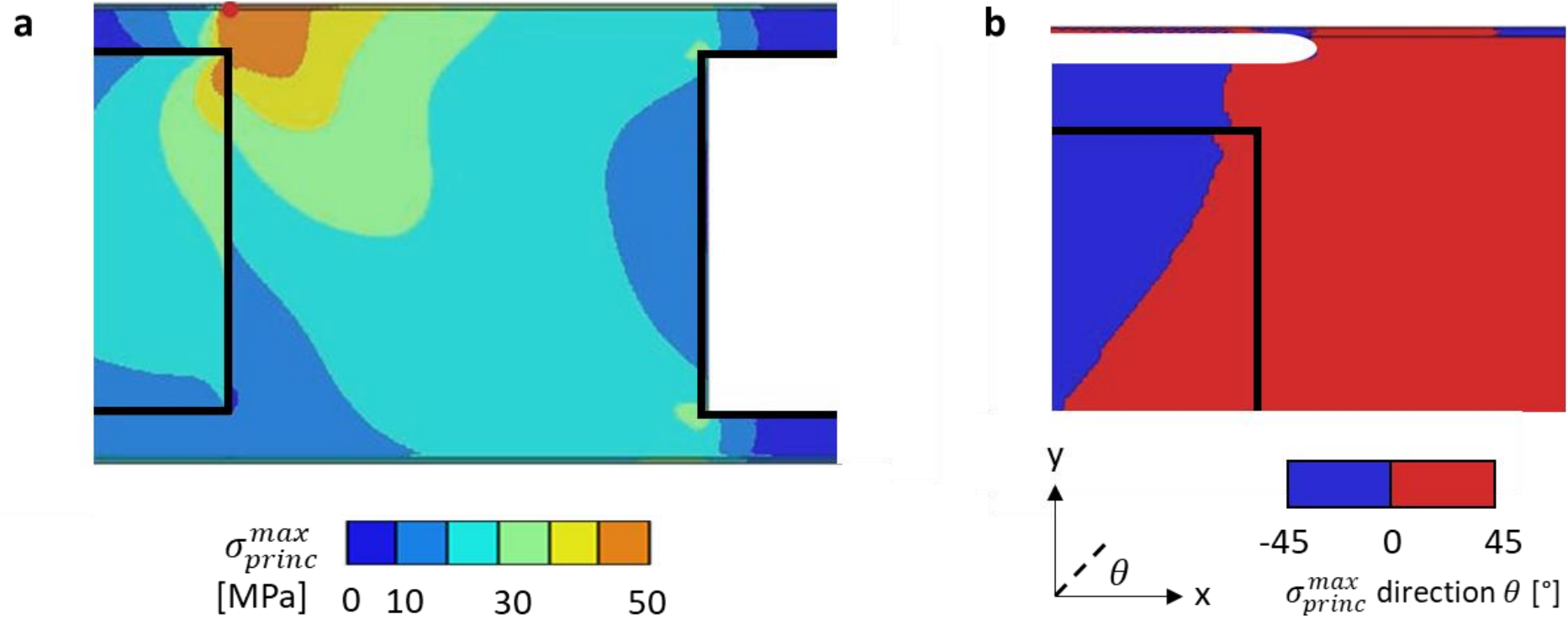
2) Secondary crack growth

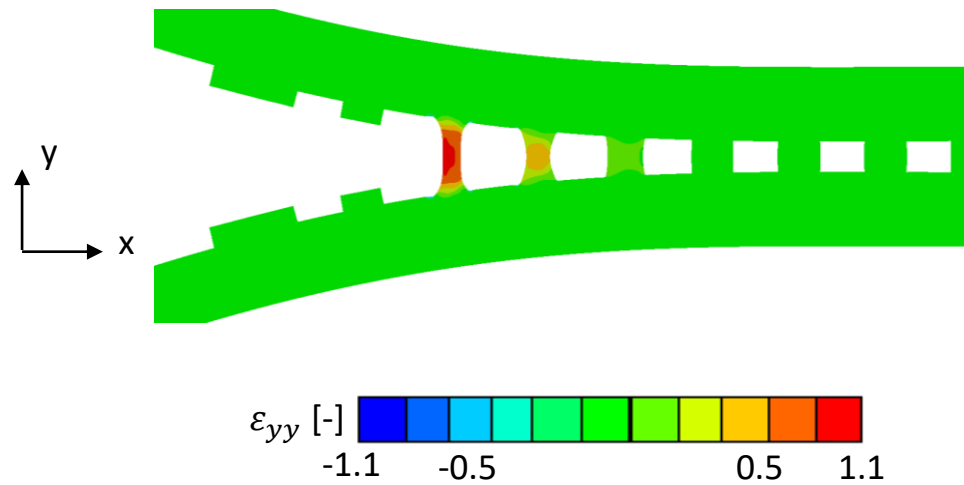


3) Crack interactions and main crack growth



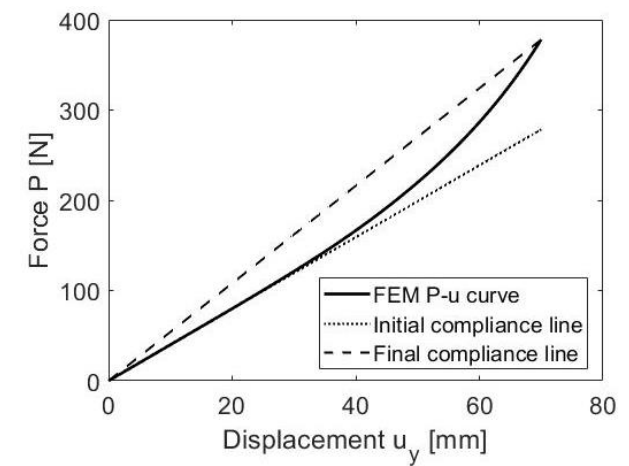
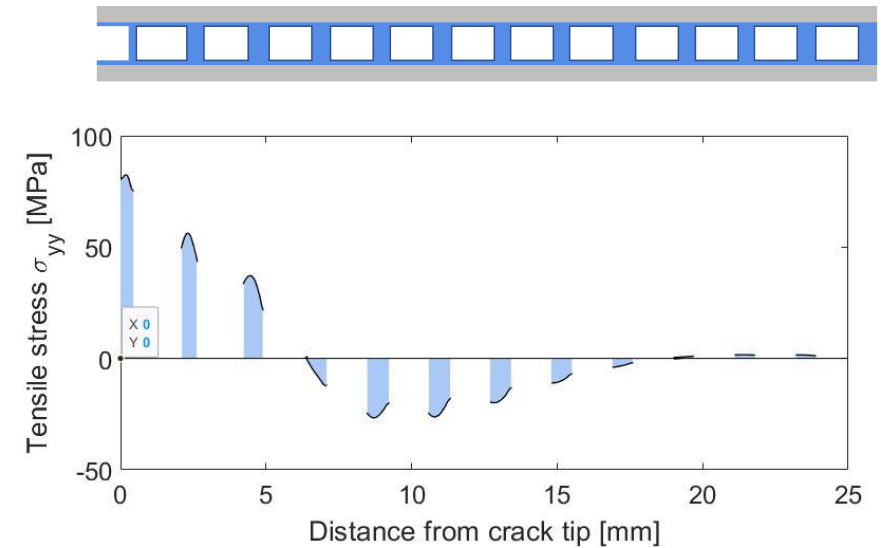
Composite Nylon Epoxy resin Cavity Propagation direction
Crack tip Secondary crack





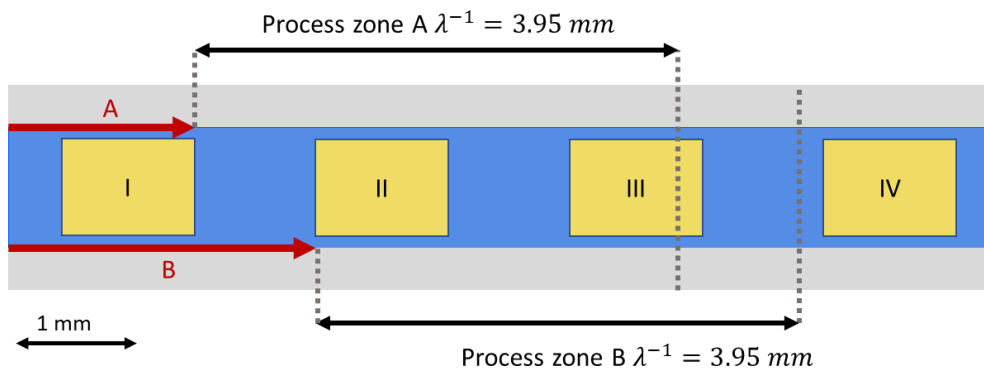
Load
redistribution

Stiffening

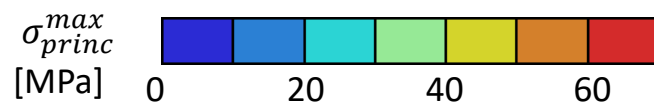
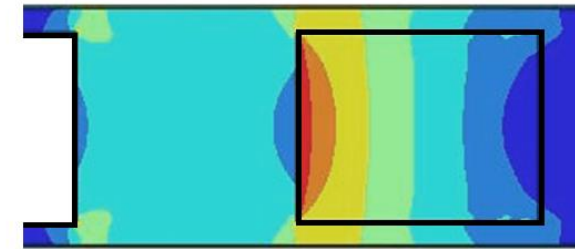
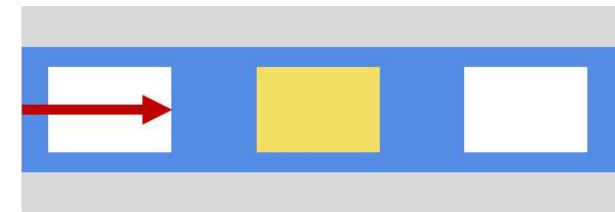
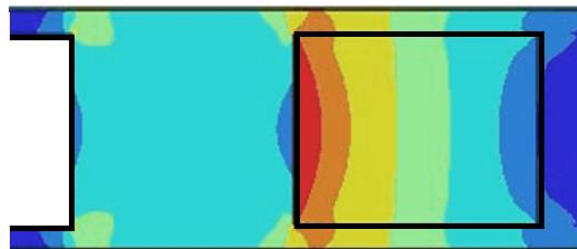
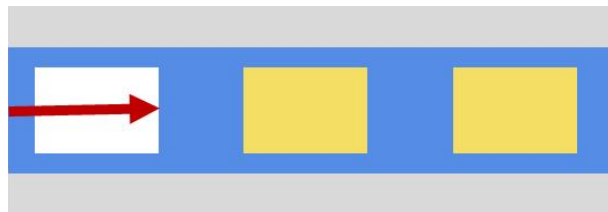


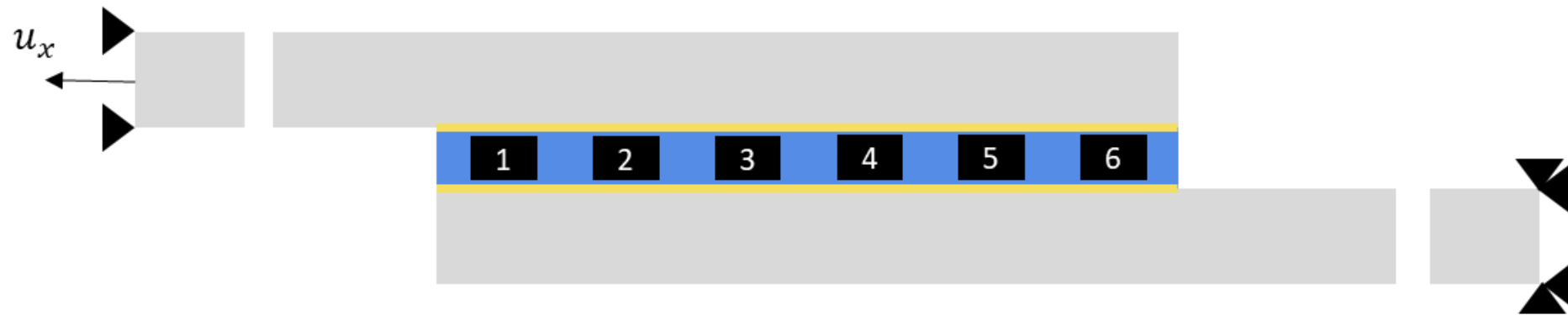
[Heide-Jorgensen et al., JMPS, 2020]

[Fulco et al., Extreme Mech. Let., 2023]



Shielding effect if III is empty





■ Composite ■ Nylon ■ Epoxy resin ■ Cavity (F = filled, E=empty)

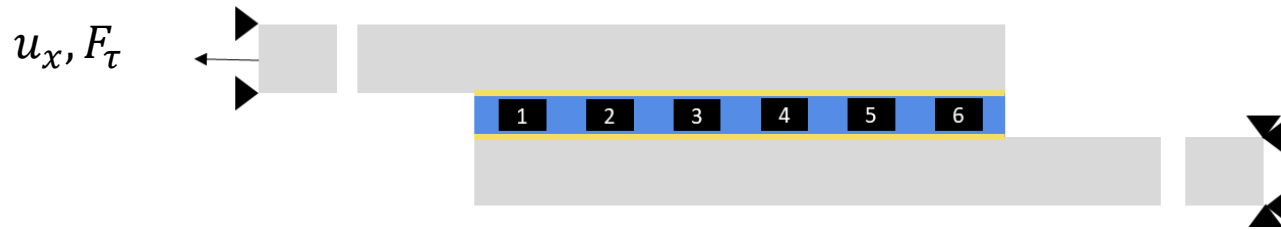
$\uparrow \tau^f, \mu$
 $\downarrow G_{Ic}$

Amount of filled cavity	Configuration	Apparent shear modulus [MPa]	Strength [MPa]	Maximum G_{Ic} [kJ/m ²]
0%	EEE	52	6	40
33%	FEE	138	9	16
	EFE	138	11	16
	EEF	137	12	16
66%	EFF	221	14	2.2
	FFE	222	12	2.2
	FEF	223	13	2.2
100%	FFF	306	14	1.5
Nylon	/	143	12	8.5 – 11.5
Resin	/	1070	20	0.09

Increasing the amount of filled channels increases shear stiffness and strength, but decreases G_{Ic}

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Nylon	/	143	12	8.5 – 11.5
Resin	/	1070	20	0.09

An empty channel located at the joint edge increases the shear strength



Large peel stresses above the filled channel
➔ More detrimental

Larger shear stresses if the channel is
empty

Advantages	Disadvantages
Homogeneous stress transfer	Surface treatment dependence
Fatigue resistance (NA)	Harsh environments (NA)
Lightweight	Lack of non-destructive control
Impact resistance (NA)	Difficult to rework
Bonding of complex shapes	Slow process
Bonding of dissimilar materials (NA)	Toxic components
Durable (NA)	Low G_{IC}
High throughput	
Decrease of cost	

Further improved by empty cavities

Probably

Flat film → issues when big t differences

Single step process instead of multiple steps process

Not needed

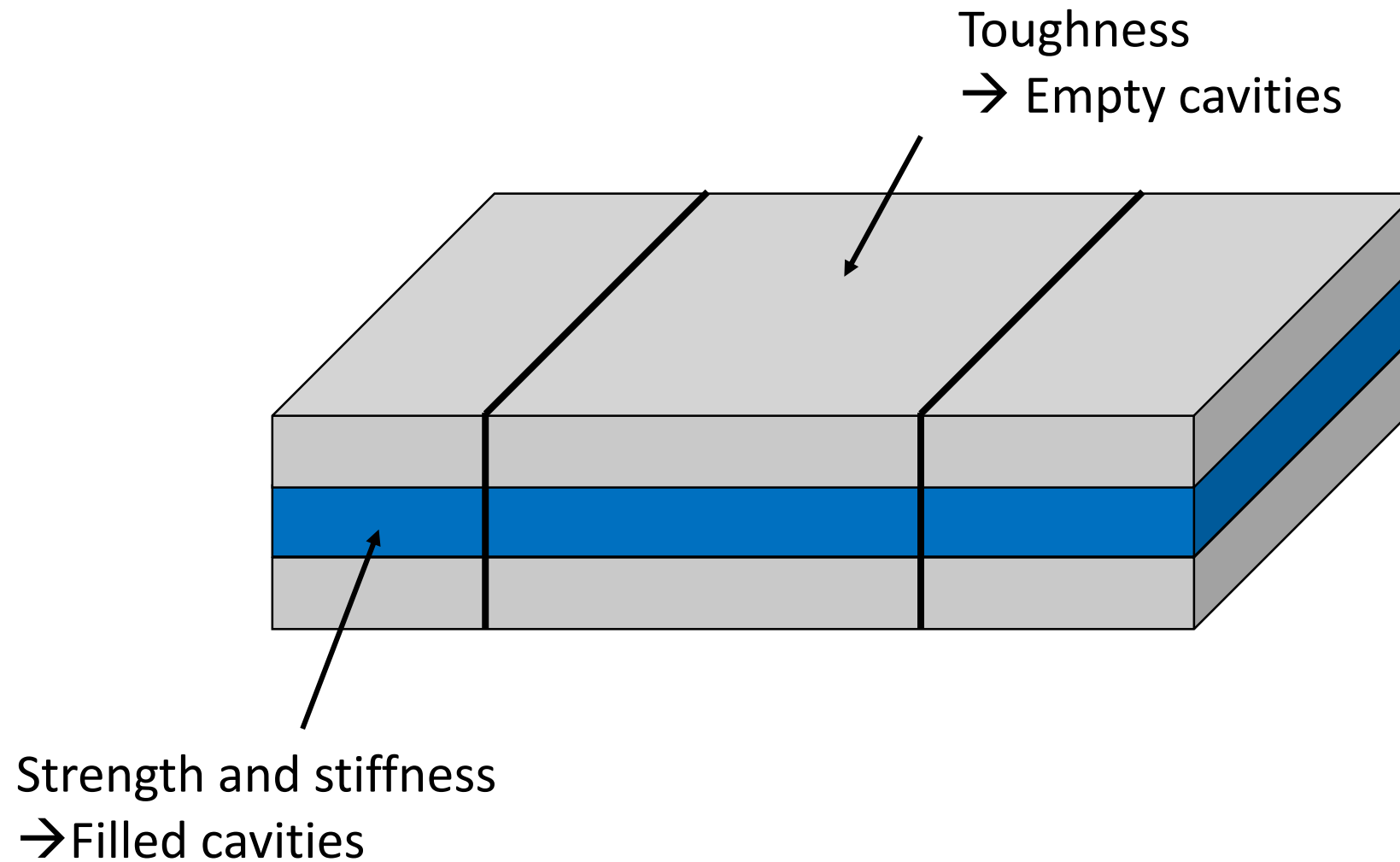
If cavity is detected → crack arrest with high G_{IC}

Single step process

Toughening

- Local control of properties by controlling the filling of the channels
- Application with $\neq$ requirements

- Integrated manufacturing: new strategy
- Need for potential applications



Tuning of the joint local properties