

Defect Engineering in *n*-Type Oxide Semiconductor TFTs

Guoli Li*, Jiawei He**, Huiru Wang*, Denis Flandre***, Lei Liao*, **

*School of Physics and Electronics, Hunan University, Changsha, China. Email: liaolei@whu.edu.cn

**School of Physics and Technology, Wuhan University, Wuhan, China

***The ICTEAM Institute, Université catholique de Louvain, Louvain-la-Neuve, Belgium

Disordered structural network and high-density intrinsic defects in amorphous oxide semiconductor materials, largely affect electrical performances (e.g. mobility, stability) of the TFTs and limit their applications in next-generation electronic device platforms, e.g. active-matrix liquid crystal display.

In our recent works, we comprehensively analyze the intrinsic defects inside these *n*-type oxide semiconductors and therefore understand the material physics to optimize the device performances via defect engineering. [1] [2] [3] [4]

In InGaZnO, besides the interface and bulk tail states, several native point defects need to be considered, as depicted in Figure 1. Regarding the neutral (Vo^0), $2+$ (Vo^{2+}) charge states related to oxygen vacancy, Vo^0 defect is known to act as deep donor, with energy level above the E_v . Electron capture takes place for the Vo^{2+} , i.e. acting as an acceptor. In InGaZnO, zinc vacancy also easily forms as deep acceptor. [3]

In ZnO, the dominant native defects are oxygen and zinc vacancies. The zinc vacancy $\text{Zn}(\text{vac})$ is deep acceptor and largely degrades the subthreshold slope SS , whereas oxygen vacancies $\text{O}(\text{vac})$ behave as deep donors and are mostly neutral and electrically inactive. In the *n*-type ZnO, zinc interstitial $\text{Zn}(\text{int})$ is strongly suggested as one dominant native shallow donor, and can easily get positively charged, resulting in negative shift of transfer curves in the pristine ZnO TFT. [1]

In In_2O_3 , the origin of the native donor is indium interstitial $\text{In}(\text{int})$, but also with the absolute coexistence of the oxygen vacancy for carrier generation. The $\text{In}(\text{int})$ as one dominant defect, can generate a shallow donor level close to the CBM. Free conduction electrons in In_2O_3 can be supplied by the $\text{In}(\text{int})$, consequently the $\text{In}(\text{int})$ is ionized to be positive univalent. On the contrary, the indium vacancies $\text{In}(\text{vac})$ form shallow acceptor levels above the VBM. [1]

While implementing the dopants incorporation and the bilayer stack, [3] [4] the defects density inside these *n*-type oxide

semiconductors can be reduced and a barrier height is formed between the hydrogenated and pristine layers, improving the device performances like the current on/off ratio, the field-effect mobility, SS and stability. However, with the continuous H plasma treatment, the H dopants can behave as the positive or negative charge states, leading to various shifting trend in the transfer curves and also device degradation. [1]

Defect self-compensation $(\text{In})^+ + (\text{O})^- \rightarrow \text{In-O}$ is also revealed to contribute to the high mobility of the bilayer-stacked InGaZnO/ In_2O_3 TFT, [2] which was deposited via a room-temperature RF sputtering technique. As schematically illustrated below, the abundant oxygen interstitials (O_i) in InGaZnO interact with the indium atoms or cluster (In-In) with dangling bonds in In_2O_3 , replacing the In-In bond by two In-O bonds as the final state.

References

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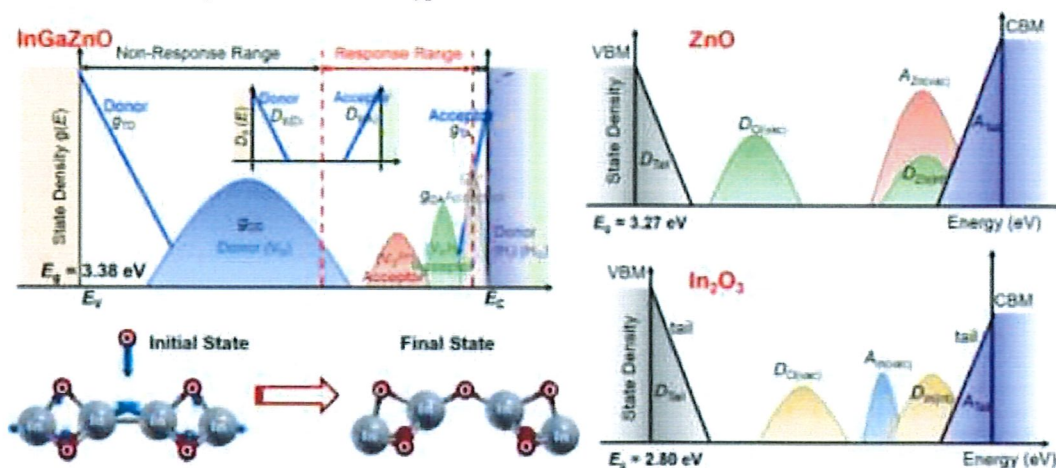


Figure 1. Schematic illustration of intrinsic defects inside the amorphous InGaZnO (with H dopants incorporation), the pristine ZnO and In_2O_3 material systems, with initial and final states of the intervention between oxygen interstitial O_i and metal cluster In-In